Outlines

- History of Conjugated Polymers
- Electronic Structures of Conjugated Polymers
- **Polymer Light-emitting Diodes**
- **Polymer-based Thin Film Transistors**
- Polymer-based Photovoltaics
- **Polymers for Memory devices**

Device Applications of Donor-Acceptor Conjugated Polymers in My Group



Polymer Thin Film Transistors

Polymer Memory Devices

Humanity's core problems in 2050





2003	6.3	Billion People
2050	8-10	Billion People

Source: Richard Smalley, Energy & Nanotechnology Conference, Houston; Christoph Brabec; Konarka .

能源之種類

- 傳統能源:
 - 石化燃料(煤[230]、石油[45]、天然氣[60])
 - 火力
 - 核能[鈾:75] 註:[]為預估使用年限,單位--年
- 再生能源:
 - 太陽能(太陽光發電、太陽熱能)
 - 風力發電
 - 生質能(廢棄物發電、沼氣發電、生質物轉化、生質物汽化發電.....)
 - 地熱發電
 - 小水力發電
 - 海洋能(波浪發電、潮汐發電、海洋溫差)



Hydroelectric 1.2 TW technically feasible

0.6 TW installed capacity

Renewable energies







Energy gap ~ 14 TW by 2050 ~ 33 TW by 2100

Credit: Christoph Brabec, Konarka

Solar 1 x 10⁵ TW at Earth surface 10,000 TW (technical value)



太陽光發電(光伏發電, Photovoltaic, PV)之特點

1. 太陽電池通常為一種固態半導體元件,將光能直接

轉換為(直流)電能,但本身不儲存能量。

2. 太陽電池使用方便、無廢棄物、無污染、無轉動部

份、無噪音、可阻隔輻射熱、或可設計為半透光。

- 3. 太陽電池模板壽命長久, 可達二十年以上。
- 4. 太陽電池外型尺寸可隨意變化,應用廣泛(小至消費

性產品--如計算機,大至發電廠皆實用)。

5. 太陽電池未來與建築物結合,將可普及化。

太陽能是免費、取之不盡、用之不竭之潔淨能源,但必須找到使用之方法。

Future Trend-- Renewable Energy

太陽光熱發電將是未來電源供應主力



、陽每天照射到地表的能量,超過全人類30年能源需求!

太陽能發電之重要發展歷史

- ✤ 1954年 Bell Labs發展出砂太陽電池
 - (Chapin等人,轉換效率約6%)
- ≥ 1958年開始太空應用(GaAs)



- ▶ 1970年開始太陽光發電系統地面應用(Si) (能源危機)
 ▶ 1976年Carlson製作出第一個非晶薄膜太陽電池
 ▶ 1980年消費性薄膜太陽電池應用(a-Si, CdS/CdTe)
 ▶ 1990年與公用電力併聯之太陽光發電系統技術成熟 (Grid-Connected PV System, Si) (電力電子技術)
 ▶ 1992年起歐美、日各國推動PV補助獎勵
- ≫ 2000年建材一體型太陽電池應用(BIPV)



History of Organic Solar Cells

	\wedge	Some important milestones in the development of organic solar cells
4		7
	2001	 Ramos used double-cable polymers in PV cells.
	2001	-Schmidt-Mende made a self-organised liquid crystalline solar cell of
		hexabenzocoronene and perylene.
	2000	-Peters / van Hal used oligomer-C60 dyads/triads as the active material
		in PV cells.
	1995	-Yu / Hall made the first bulk polymer/polymer heterojunction PV.
	1994	-Yu made the first bulk polymer/C ₆₀ heterojunction PV.
	1993	-Sariciftci made the first polymer/C ₆₀ heterojunction device.
	1991	-Hiramoto made the first dye/dye bulk heterojunction PV by co-sublimation.
	1986	- Tang published the first heterojunction PV device.
	1964	-Delacote observed a rectifying effect when magnesium phthalocyanines
		(CuPh) was placed between two different metalelectrodes.
	1958	-Kearns and Calvin worked with magnesium phthalocyanines (MgPh),
		measuring a photovoltage of 200 mV.
	1906	-Pochettino studied the photoconductivity of anthracene.
	1839	-Becquerel observed the photoelectrochemical process.
- 1		

太陽能發電(Photovoltaic)原理



•太陽電池是以P型與N
型半導體材料接合構成的
裝置。
• 當陽光照射太陽電池
時,半導體材料吸收光子
會產生電子-電洞對。
• 分離正電荷(Hole)、負電
荷(Electron)會分別往正(P
型)、負(N型)極方向移動
並且聚集。
• 正、負極接上負載時,
將有電流流出,可以對負
載作功(燈泡會亮、馬達會
轉)。

太陽能電池種類



Highest Solar Cell Efficiency Tables until 2006. 04

Table I. Confirmed terrestrial cell and submodule efficiencies measured under the global AM1.5 spectrum (1000 Wm^{-2}) at 25°C

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Classification ^a	Effic. ^b (%)	Area ^c (cm2)	V _{oc} (V)	J _{sc} (mA/cm ²)	FF ^d (%)	Test Centre ^e (and Date)	Description
Silicon							
Si (crystalline)	24.7 ± 0.5	4.00 (da)	0.706	42.2	82.8	Sandia (3/99)	UNSW PERL ¹⁴
Si (multicrystalline)	20.3 ± 0.5	1.002 (ap)	0.664	37.7	80.9	NREL (5/04)	FhG-ISE ¹⁵
Si (thin film transfer)	16.6 ± 0.4	4.017 (ap)	0.645	32.8	78.2	FhG-ISE (7/01)	U. Stuttgart (45 µm thick) ¹⁶
Si (thin film submodule)	9.4 ± 0.3	94.9 (ap)	0.493 ^f	26.0^{f}	73.1	Sandia (4/06)	CSG Solar $(1-2 \mu m \text{ on glass};$ 20 cells) ⁵
II–V Cells							
GaAs (crystalline)	25.1 ± 0.8	3.91 (t)	1.022	28.2	87.1	NREL (3/90)	Kopin, AlGaAs window ¹⁷
GaAs (thin film)	24.5 ± 0.5	1.002 (t)	1.029	28.8	82.5	FhG-ISE (5/05)	Radboud U., NL ¹⁸
GaAs (multicrystalline)	18.2 ± 0.5	4.011 (t)	0.994	23.0	79.7	NREL (11/95)	RTI, Ge substrate ¹⁹
InP (crystalline)	21.9 ± 0.7	4.02 (t)	0.878	29.3	85.4	NREL (4/90)	Spire, epitaxial ²⁰
Thin Film Chalcogenide							
CIGS (cell)	$18.4\pm0.5^{\rm g}$	1.04 (ap)	0.669	35.7	77.0	NREL (2/01)	NREL, CIGS on glass ²¹
CIGS (submodule)	16.6 ± 0.4	16.0 (ap)	0.661^{f}	33.4 ^f	75.1	FhG-ISE (3/00)	U. Uppsala, 4 serial cells ²²
CdTe (cell)	$16.5\pm0.5^{\rm g}$	1.032 (ap)	0.845	25.9	75.5	NREL (9/01)	NREL, mesa on glass ²³
Amorphous/Nanocrystalline Si							
Si (amorphous) ^h	9.5 ± 0.3	1.070 (ap)	0.859	17.5	63.0	NREL (4/03)	U. Neuchatel ²⁴
Si (nanocrystalline)	10.1 ± 0.2	1.199 (ap)	0.539	24.4	76.6	JQA (12/97)	Kaneka $(2 \mu m \text{ on glass})^{25}$
Photochemical							
Dye sensitised	10.4 ± 0.3	1.004 (ap)	0.729	21.8	65.2	AIST (8/05)	Sharp ²⁶
Dye sensitised (submodule)	6.3 ± 0.2	26.5 (ap)	6.145	1.70	60.4	AIST (8/05)	Sharp ⁶
Organic							
Organic polymer ⁱ	3.0 ± 0.1	1.001 (ap)	0.538	9.68	52.4	AIST (3/06)	Sharp, fullerene derivative ⁷
Multijunction Devices							
GaInP/GaAs/Ge	32.0 ± 1.5	3.989 (t)	2.622	14.37	85.0	NREL (1/03)	Spectrolab (monolithic)
GaInP/GaAs	30.3	4.0 (t)	2.488	14.22	85.6	JQA (4/96)	Japan Energy (monolithic) ²⁷
GaAs/CIS (thin film)	25.8 ± 1.3	4.00 (t)	-	_	-	NREL (11/89)	Kopin/Boeing (4 terminal) ²⁸
a-Si/μc-Si (thin submodule) ⁱ	11.7 ± 0.4	14.23 (ap)	5.462	2.99	71.3	AIST (9/04)	Kaneka (thin film) ²⁹

Prog. Photovolt: Res. Appl. 2006; 14:455-461

Efficiency of Photovoltaic Devices

Measured Solar Cell Efficiency from 1975 to the present



From NREL, USA

Requirement Trends of the Photovoltaic Industry



Annual PV cell/module production

Estimation of present and future installed solar capacities

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太陽光發電之應用領域



Solid State Organic Solar Cells

	Optical power \Rightarrow electrical power
Photovoltaics -	FIGURES OF MERRIT:
	Power conversion efficiency
	Full solar intensities
	Reliability

High absorption in the visible spectrum

Have relaxed deposition requirements

can be manufactured in a low cost process

can be grown on thin and flexible substrate

can add value to existing product

Challenge!

Current power conversion efficiencies are too low for commercial implementation

有機太陽光電技術

Organic solar cells : three types





Organic double layer p/n cell



C. W. Tang, Appl. Phys. Lett. 1985, 48, 183.

Bulk heterojunction solar cells

Charge separation in nanostructured composite organic semiconductors





nanoscopic mixing of donor and acceptor to overcome ~10 nm exciton diffusion length

R. H. Friend et al., Nature 1995, **376**, 498 *A. J. Heeger et al., Science 1995,* **270**, 1789



Structures and Principles of LED & Photovoltaic mode

LED mode

Photovoltaic mode

A PV mode is the reverse of a LED. In PVs electrons are collected at the metal electrode and holes are collected at the ITO electrode.

What & Why Is Organic Solid Phase Photovolta

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Polymer Based Solar Cell-Active layer



Light Converting Processes



General Mechanism in Organic Photovoltaic Cells



- (1) Photon absoption (η $_{\rm A})$
- (2) Generation of excitons
- (3) Exciton diffusion (η_{diff})

(4) Hole-electron separation (η _{TC})

(5) Carrier transport towards the electrode ($\eta_{\rm tr})$

(6) Charge collection at the respective electrode(η _{cc})

General Scheme for Organic Photovoltaic Effect



Examples on Polymer Phtovoltaic Devices



First Polymer-Polymer heterojunction PV

NATURE · VOL 376 · 10 AUGUST 1995



Energy conversion efficiency: 2.9 % Science 1995

Plastic Solar Cells



A large area plastic solar cell running a small motor



MDMO-PPV:PCBM solar cell realized on a PET substrate

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Applications



Converse to 2006 ITDI - 要社保证依障

Polymer solution processed cells come in three 'flavors'



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Polymer solar cells





Converse to 2006 ITPI + 坐住 在在地

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建構高分子太陽能電池元件製程平台



標準元件製作流程

Procedure:



太陽電池之 I-V曲線



ill Factor (F.F.) = (V_{mp} x I_{mp} / V_{oc} x I_{sc}) x 100% **、陽電池效率(Efficiency**; η) = (I_{sc} x V_{oc} x FF / 輸入日照功率) x 100%
輸入日照功率(W)=太陽電池面積(m²) ×日照強度(W/m²)
日照強度為1000 W/m²之最大輸出功率即為Wp
太陽電池開路、短路時皆不會燒燬



Air Mass (AM)- A measure of how much atmosphere sunlight must travel through to reach surface. The intensity is fixed at 100W/cm².

Open circuit voltage (V_{oc}) Voltage across the cell in sunlight when no current is flowing.

Short circuit voltage (I_{sc}) Current flows through an solar cell when there is no external resistance.

Maximum power point (mpp) The maximum power is produced.

Fill Factor (FF) $FF = \frac{I_{mpp}V_{mpp}}{I_{sc}V_{oc}}$ Power conversion $\eta_e - \frac{I_{mpp}V_{mpp}}{P_{in}} - \frac{I_{sc}V_{oc}FF}{P_{in}}$

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Polymer Photovoltaic Structure



Postproduction induced P3HT:PCBM solar cells

P3HT:PCBM solar cells

	AM1.5 performance							
$J_{ m sc}$	$V_{ m oc}$	FF	EQE	η				
8.7	0.58	0.55	70	2.8				
8.5*	0.55	0.60	70	3.5				
9.4	0.61	0.53	58	3.0				
7.2	0.62	0.62	58	2.7				
11.1*	0.65	0.54		4.9				
9.5*	0.63	0.68	-	5.0				
10.6	0.61	0.67	63	4.4				
> 10**	~0.60	-	73	4.4				

* measured at 80 mW/cm²; ** at 85 mW/cm²

- P. Schilinsky et al., Appl. Phys. Lett. 2002, 81, 3885.
- F. Padinger et al., Adv. Funct. Mater 2003, 13, 85.
- Y. Kim et al., Appl. Phys. Lett. 2005, 86, 063502.
- X. Yang et al., Nano Lett. 2005, 5, 579.
- M. Reyes-Reyes et al., Appl. Phys. Lett. 2005 87, 083506
- W. Ma et al., Adv. Funct. Mater., 2005, **15**, 1617.
- G. Li et al., Nature Mater. 2005, 4, 864.
- Y. Kim et al., Nature Mater. 2006, 5, 197.







Yang Yang et al. Nature Mater. 2005, 4, 864.

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R&D Focuses to Tackle Critical Issues

- New polymer design to enhance mobility & light harvesting
- Ordered BHJ structure with well-defined paths and limited width
- New device design to enhance Jsc or Voc
- Life time issues





Current Challenges

The lower photocurrent is due to poor light absorption, generation and transport. The fill factor is due to poor transport and recombination.

• Improving light harvesting

Small band gap polymer, dye-sensitized materials, light-trapping structures

Improving charge transport

Carrier mobility (10⁻²~10⁻⁵ cm²/VS) is low

Control morphology

Processing condition, self organization, synthesis of D-A block copolymer, use of porous films as template

• Addressing manufacturing issue and improving stability

By encapsulating cells and more stable materials

• Understanding device function and limits to performance

國際發展比較-有機固態太陽光電

高分子-碳材太陽光電池技術								
Authors/Institutes	Compositions	Efficiency (%)						
David L. Carroll/Wake Forest University, USA/Kyungkon Kim/ KIST, Korea	ITO/PEDOT:PSS/P3HT:PCBM/LiF/AI	6.1						
C. J. Brabec /Siemens (Konarka) AU/A. Heeger/Konarka USA	ITO/PEDOT:PSS/P3HT:PCBM/AI	5.68						
Kwanghee Lee /Pusan U., Korea/ A. Heeger	ITO/PEDOT:PSS/P3HT:PCBM/TiO2/Al	5.8						
A. Heeger/UC Santa Barbara , USA	ITO/PEDOT:PSS/P3HT:PCBM/AI	4.8-5.1/100 samples						
N. S. Sariciftci /Linz Austria	ITO/PEDOT:PSS/MDMO- PPV:PCBM/LiF/Al	3.5						
Y. Yang /UCLA ,USA	ITO/PEDOT:PSS/P3HT:PCBM/AI	4.4/3.8						
R. A. J. Janssen/ECN Netherlands	ITO/PEDOT:PSS/MDMO- PPV:[70]PCBM/LiF/Al	3.0						
ITRI	ITO/PEDOT:PSS/P3HT:PCBM/Ca:Al	5.4						





Prospects for high-efficiency (>10%) Polymer PV cells

- **1. New device designs: Ordered Bulk Heterojuctions**
- Approach: Polymer Semiconductor/acceptor Order Heterojuction Structure
- 2. The high light-absorbing capabilities: Conjugated polymer and electron acceptor with lower band gap: 350-900 nm (3.5~1.4eV)
 - Approach: Low Eg Polymer with a broad absorption
- **3. Suitable energy level for high Voc: Low HOMO level**
- 4. Higher carrier mobilities:
 - Approach :高分子mobilities > 0.01 cm²/Vs and Morphology control
- 5. Optimized Processing Conditions: coating/drying/additives

Four device architectures of conjugated polymer–based PV cells



- Single Layer(a): Low EQE(0.1~1%) due to exciton recombination; low carrier mobility
- Bilayers (b) : PA-PPV/TiO₂ 25% EQE, 3.9 % power efficiency (435 nm);

PPV/BBL 66%% EQE, 2% power efficiency

• Bulk heterojunction (c) (d):

PPV/C60 Derivatives 70% EQE, 3.5% power efficiency

Chem. Mater. 2004, 16, 4533

Organic Photovoltaic Device Architectures



Organic Photovoltaic Device Architectures

Bulk Heterojunction Devices



Organic Semiconductors Used in Solar Cells











Absorption Spectrum of Organic Materials



Photon reflux from the sun (AM 1.5)

Design strategy for low band gap: Donor-Acceptor polymers

increase the double bond character of the single bonds:



D-A Conjugated Alternating Polymers: PCBM Solar Cells





- $V_{oc} = 0.72 V$ FF = 0.37
- $I_{sc} = 3.1 \text{ mA/cm}^2$ PCE = 1 %

 $V_{oc} = 0.76 V$ FF = 0.49 $I_{sc} = 4.31 \text{ mA/cm}^2$ PCE = 1.6 %



 $V_{oc} = 0.72 V$ FF = 0.46 $I_{sc} = 4.66 \text{ mA/cm}^2$ PCE = 2.2 %



 $V_{oc} = 0.56 V$ FF = 0.49 $I_{sc} = 3.6 \text{ mA/cm}^2$ PCE = 0.51 %

D-A Conjugated Alternating Polymers: PCBM or C70 Solar Cells



D-A Conjugated Alternating Polymers: PCBM Solar Cells



 $I_{sc} = 3.1 \text{ mA/cm}^2$

FF = 0.37

PCE = 1 %



 $V_{\rm oc} = 0.77 \ V$

 $I_{sc} = 3.4 \text{ mA/cm}^2$

FF = 0.42

PCE = 0.2 %



 $V_{oc} = 0.59 V$ FF = 0.39 $I_{sc} = 2.6 \text{ mA/cm}^2$ PCE = 0.6 %



 $V_{\rm oc} = 0.56 V$

 $I_{sc} = 3.5 \text{ mA/cm}^2$

FF = 0.58

PCE = 1.1 %



 $V_{oc} = 0.61 V$ FF = 0.24 $I_{sc} = 0.2 \text{ mA/cm}^2$ PCE = 0.02 %



Table 1. Molecular Weights, FET Mobility, and Optical and Electrochemical Properties of Various Polymers

	$M_{\rm W}~({ m FDI})$	λ_{max} (film)	α^a (cm ⁻¹)	E_8^{opt} (eV)	E_{ox} (V) ^b	IP (eV) (HOMO)	EA (oV) (LUMO)	μ_{h}^{c} (cm	2/(V s))	on/off ^c
P1	25200 (1.52)	490	6.3×10^{5}	2.10	0.71	5.17	3.07	3.7×10^{-4}		5.2×10^{3}
P2	21300 (1.97)	510	3.0 × 10 ⁵	2.10	0.72	5.18	3.08	1.5×10^{-4}		1.4×10^{4}
P3	48700 (2.19)	510	9.9 × 10 ⁵	2.08	0.64	5.1	3.02	8.3×10^{-4}		9.1×10^{4}
P4	29300 (1.87)	508	9.4×10^{5}	2.11	0.72	5.18	3.07	3.0×10^{-3} ($9.9 \times 10^{-4})^{d}$	1.3 × 10 ⁶ (6.5 × 10 ⁵) ^d
P3HT	47000 (2.45)	552	1.2×10^{6}	1.90	0.74	5.20	3.30	6.5×10^{-2}	1.5	1.3×10^{3}

^{*a*} Absorption coefficient was determined at λ_{max} in THF. ^{*b*} E_{ox} is the onset potential of oxidation of polymer. ^{*c*} Thin-film FETs were fabricated from 1 wt % *o*-DCB solutions. ^{*d*} CHCl₃ solvent was used instead of *o*-DCB.

High absorption coefficient in comparison to P3HT

Ko et al., Macromolecules **2008**, 41, 5519

Table 1. Molecular Weights, OTFT Mobility, Optical and Redox Properties of Various Polymers

	M _w (PDI)	λ_{\max} (film)	$\alpha(\times10^{5}~\text{cm}^{-1})$	Eg ^{opt} (eV)	E _{ox} °(V)	HOMO (eV)	LUMO (eV)	$\mu_{\rm h}~({\rm cm^2/Vs})$	on/off
P1	26300 (1.55)	520	$\frac{11^a(1.6)^b}{7.7^{a}(1.4)^b}$	1.76	1.00	-5.46	-3.56	7.0×10^{-4}	1.3×10^{5}
P2	38600 (1.74)	590		1.70	0.97	-5.43	-3.66	3.4×10^{-3}	5.6×10^{6}

Ko et al, J. Am. Chem. Soc. 2008, in press

Design Rules for Donors in Solar Cell - Towards 10 % PCE

To get PCE >10%The highest PCE of polymer solar cell up to now : 7.9%Bandgap of donor polymer < 1.74 eV & LUMO < -3.92 eV</th>

PV efficiency

The importance of HOMO/LUMO level on the Photovoltaic Devices

4.00

5.00 6.00

7 00

8 00

10.00

LUMO -4.3 eV

HOMO -6.0 eV

PCBM

Scharber et al., Adv. Mater. 2006, 18, 789

NewQuinoxaline based Donor-Acceptor Conjugated PolymersFor Optoelectronic Applications

Highest Solar Cell PCE :1.76%

J. Polym. Sci. Polym. Chem. 2009, Through the collaboration of W. C. Chen and F. C. Chen

 M_n =22308 PDI=2.38 HOMO= -5.18 eV LUMO= -3.60 eV Mobility 9.25x10⁻⁴ On/off 2.31x10⁴

M_n=4003 PDI=1.26

HOMO= -5.06 eV LUMO= -3.36 eV Mobility 9.25x10⁻⁴ On/off 2.3x10⁴

 M_n =8590 PDI=1.67 HOMO= -4.91 eV LUMO= -3.03 eV Mobility 4.71x10⁻⁵ On/off 4.07x10³

Mn=8450 PDI=1.76 HOMO= -5.06 eV LUMO= -3.19 eV Mobility 2.52x10⁻⁴ On/off 2.00x10⁴

4.8 6.3 7.4

				SC		-	
						Best	
PBDTTT-E	-3.24	-5.01	0.62	-13.2	63	5.15	
PBDTTT-C	-3.35	-5.12	0.7	-14.7	64.1	6.58	
PBDTTT-CF	-3.45	-5.22	0.76 🔻	-15.2	66.9	7.73	

Bulk heterojunction solar cells with internal quantum efficiency approaching 100%

YU, L. Nat. Photon. 2009, 3, 297.

With TiO_x layer, the IQE, EQE, absorption could be enhanced, leading to the best efficiency up to 6.1%

Some of the Approaches in MCL

Better packing conducting polymer

- Stereo-regular conducting polymer (no micro-structure defects)

- Tightly-packing conducting polymer but still maintain its processbility
- High mobility conducting polymer design with good solubility

P3HT:PCBM Solar Cells

Different annealing time

 $V_{oc} = 0.61 V$ FF = 0.67

I_{sc} = 10.6 mA/cm² PCE = 4.37 %

PPV:PCBM Solar Cells

 $V_{\rm oc} = 0.82 \ V$ FF = 0.61

I_{sc} = 5.25 mA/cm² PCE = 2.5 %

Importance of Polymer Morphology on Photovoltaic Efficiency

Figure 2. The AFM height (a–d) and simultaneously taken phase (e–h) images of the MDMO-PPV/PCBM composite films of 90 (a,e), 80 (b,f), 67 (c,g), and 50 wt-% PCBM (d,l). Height bar (maximum peak-to-valley) represents 20 nm (a), 10 nm (b), 3 nm (c), and 3 nm (d). The size of the images is 2.0 μ m x2.0 μ m.

Morphology determining parameters: The spin casting solvent The composition between polymer and fullerene The solution concentration The controlled phase separation and crystallization induced by thermal annealing The chemical structure of the materials

Adv. Funct. Mater. 2004, 14, No. 5, May J. Mater. Chem., 2006, 16, 45-61 H. Spanggeard, F.C. Krebs / Solar Energy Materials & Solar Cells 83 (2004) 125-146

Enhanced thermal stability and efficiency by Low Temp. Drying

With the LT process, the low drying temperature <u>enhanced</u> <u>P3HT nucleation</u>, but <u>restricted P3HT/PCBM phase</u> <u>separation</u> (due to the limited movements of the molecules under the low temperature), resulting in a morphology where abundant P3HT nuclei dispersed within a fine mixture of PCBM-rich domains and amorphous P3HT-rich domains.

Adv. Funct. Mater. 2010, 20, 834

Figure 7. Schematic representations illustrating the morphology of the active layer upon drying (top) and annealing (bottom): a) the RT and b) LT layers.

Figure 6.	UV-vis	spectra	of the	RT	and LT	layers	(annealed	at 190°C for	
2 min).									

Process	Annealing	J _{SC} [mA cm ⁻²]	$V_{OC}[mV]$	FF [%]	PCE [%]
RT	None	9.42 ± 0.54	534 ± 18	63.7 ± 1.6	3.20 ± 0.14
RT	2 min/190 °C	8.41 ± 0.46	587 ± 16	65.4 ± 1.3	3.23 ± 0.18
RT	30 min/190 °C	8.25 ± 0.39	566 ± 14	53.3 ± 4.6	2.49 ± 0.18
RT	8 h/100 °C	1.72 ± 0.28	545 ± 7	39.0 ± 2.1	0.36 ± 0.05
RT	1368 h/65 °C [a]	$\textbf{2.48} \pm \textbf{0.29}$	487 ± 14	37.2 ± 2.3	$\textbf{0.4.5} \pm \textbf{0.07}$
LT (-5 °C)	None	1.40 ± 0.47	418 ± 15	33.9 ± 0.9	0.20±0.06
LT (-5 °C)	2 min/190 °C	11.50 ± 0.73	615 ± 11	62.8 ± 2.4	4.43 ± 0.31


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Best PCE~4.7%
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Efficiency enhancement in low-bandgap polymer solar cells by processng with alkane dithiols Heeger, Nat. Mater. 2007, 6,497.

Defined transport path

From Polymer:

URV 35453800 300mm WD 52,3mm

From Nanocrystals

Control of Surface Grown ZnO

Inverted semi-tranparent OPVs

F. C. Chen, J. L. Wu, K. H. Hsieh, and W. C. Chen, Org. Electron. 2008

J-V characteristics

Incorporating an Al CE grid of 10% shadow fraction, the semi-transparent OPVs exhibited very similar electrical characteristics whether the device was illuminated from the bottom side (PCE = 3.15%) or from the top side (PCE = 2.8%).

高分子太陽能電池封裝技術之開發

Adding a hydrophobic layer, including a ALD HfO₂ film or a PVDC film, on top of the ALD Al₂O₃ increased the lifetime to up to 282 hrs

